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FUEL ELEMENT PRODUCTION AT THE  
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**COORDENADORIA DE METALURGIA NUCLEAR  
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# TECHNOLOGY DEVELOPMENT FOR THE POWER REACTORS FUEL ELEMENT PRODUCTION, AT THE INSTITUTO DE ENERGIA ATÔMICA – SÃO PAULO\*

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## ABSTRACT

The authors describe the work done at the Divisão de Metalurgia Nuclear of the Instituto de Energia Atômica on the technology of fuel elements fabrication.

The obtained results in sintering  $UO_2$  locally produced are discussed in relation to the problems found in the characterization of powders of this oxide. The tests used comprise chiefly the BET, Fisher and both transmission and scanning electron microscopy methods.

Details of the Nuclear Ceramics Pilot Plant of the Instituto de Energia Atômica with relation to the programs to be considered in near future are also included.

## INTRODUCTION

In the development of  $UO_2$  pellets technology aiming the fuel elements fabrication for PWR type power reactors in the future, the Divisão de Metalurgia Nuclear of the Instituto de Energia Atômica has systematically made experimental studies aiming the characterization of natural  $UO_2$  powders, its correlation with the properties of the sintered pellets and the selection of their important characteristics in the texture. The characterization of the  $UO_2$  powders produced in the Divisão de Metalurgia Nuclear from purified salts prepared in the Divisão de Engenharia Química seeks the necessary operational control and assures growing experience in this specialized technology. The elucidation of the texture aspects is mainly related with the tolerances in the pellets dimensions to be kept up and with the level of permissible rugosity at their surfaces.

Finally, the data being obtained which constitute a increasing experience in the  $UO_2$  pellet fabrication technology can be better evaluated when the results under irradiation which at present are in planning phase are known.

As said before, this task accomplished during the last eight years is oriented toward the solution of specific problems of fuel elements fabrication aiming the accumulation of an adequate know-how for future expansions with basis in the adaptation and in the changes of technology for the standing conditions.

A number of technical papers recently presented by the authors<sup>[2, 5, 13]</sup> to other Congresses both of metallurgy and of atomic energy has described details of the work done. In this paper the authors resume the main results obtained and go over the data related to the finishing characteristic of the pellets and their textures.

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(\* Presented at the III Inter American Conference on Materials Technology, August 14-17, 1972 – Rio de Janeiro, GB.

## PROPERTIES OF THE $UO_2$ POWDERS

Natural  $UO_2$  powders utilized until the present in experimental work have been produced at the Divisão de Metalurgia Nuclear from ammonium diuranate of high purity prepared under different conditions, by the Divisão de Engenharia Química. Due to close link between the two Divisions it has been possible to adjust the precipitation conditions of the salt in order to assure firmly the desired characteristics for the  $UO_2$  powders. This articulation is assuring appreciable improvements on the characteristics of that salt and of the  $UO_2$  powders resultant from the different qualities of it in order to meet the specifications which have been set up progressively.

In a initial step the process of treatment was essentially based upon the utilization of the ion exchange process in a uranyl sulphate solution. Although all the characteristics of the powders resultant from this procedure were quite satisfactory this type of ammonium diuranate retained a certain proportion of ion sulphate resulting special characteristics for the powders not only for the sintered pellets but also for the dispersion in the  $U_2O_3$  form used in the fuel element plates.

More recently this process was changed by a solvent extraction method utilizing aqueous uranyl nitrate solution.

The calcining and reducing variables mainly time, temperature, heating and cooling time rates and time at a lower temperature in the water cooled region of the hydrogen reduction furnaces determine together with the intrinsic properties of the uranium salt their physical and physico-chemical properties and as consequence their sinterability.

The uranium oxides utilized in the development of this work were prepared always from uranium salts produced in the IEA by three different conditions as follows: one called  $UO_2$  type N prepared through the precipitation of aqueous uranyl nitrate solution in a pH near seven; the  $UO_2$  type S resulting from ammonium neutralization at  $60^\circ C$  in a pH near 7. In general these salts are previously calcined to  $U_3O_8$  although this technique was not followed in the case of the  $UO_2$  type S which is prepared only by hydrogen reduction at  $650^\circ C$  during one hour. In the other cases calcining operation has been carried out at  $500^\circ C$  for three hours followed by hydrogen reduction at  $750^\circ C$  for one hour both in the cases of  $UO_2$  type N and of  $UO_2$  type D prepared from dissociated ammonia. These marked differences of origin and of treatment determine some properties too much different for the  $UO_2$  powders as it will be shown later.

## CHARACTERIZATION TESTS OF THE POWDERS OF URANIUM OXIDES

The most utilized tests are the following:

1) **Particle density** — The used process is similar to be one developed at Oak Ridge National Laboratory<sup>(14)</sup> for the determination of the density of  $UO_2$  particles. The equipment is composed mainly of a 20 ml pycnometer, thermometer, charging tube and vacuum chamber for the introduction of mercury. When  $U_3O_8$  is utilized toluene is used instead of mercury.

2) **Average particle size** — It is utilized the Fisher Sub Sieve Sizer equipment. The instrument has a device to press the powder to a certain porosity level when air is introduced. The air is supplied by a small compressor through the sample. The resultant head loss is measured by an water manometer. This equipment gives reproducible results in the range of 0.2 to  $50 \mu$ .

3) **O/U Ratio** — The utilized process for the determination of the value of the O/U ratio is analytical and based upon the determination of the contents of total U and hexavalent U respectively.

4) **Surface Area** — The values of the surface areas of the  $UO_2$  powders were determined through a CG model 10 equipment as shown in figure 1. It makes essentially use of the dynamic method for adsorption initially developed by Nelsen and Eggertsen<sup>(14)</sup> where a mixture of nitrogen and

helium flows steadily through one layer of the sample the composition of the flowing gas is determined by measurements of thermal conductivity being registered the areas corresponding to adsorption and desorption peaks in a integrator recording potentiometer. In order to calculate the saturation pressure of the nitrogen value at around minus  $196^{\circ}\text{C}$  it was used a thermometer of oxygen saturation pressure made at the Divisão de Metalurgia Nuclear with this equipment it is possible to register temperature variations of less than  $0.05^{\circ}\text{C}$  in the liquid nitrogen bath.



Figure 1 – Surface area determination apparatus used for  $\text{UO}_2$  powders

6) X Ray diffraction – In certain cases the control utilizes X ray diffraction data obtained in a Rigaku unit by the powder method

6) Optical microscopy – The examinations are done with powders dispersed in alcohol on a glass plate at the metallographic microscopy Leitz MM 5. In order to have uniform dispersions they are obtained by ultrasonics during five minutes

7) Electron microscopy – Different types of powders have been examined under electron microscope and these examinations have been done at the Centro de Microscopia Eletronica of the Instituto de Física Universidade de São Paulo in a electron microscope Siemens-Elmiskop 1 carbon replicas preshadowed with platinum are used. To study the textures of the sintered pellets the Divisão de Metalurgia Nuclear has a Cambridge Stereoscan unit model S 4. Such apparatus is being completed with equipment for a two channel X rays spectrometer and a converter connected to a microanalysis recorder

8) Differential thermal analysis – The data of differential thermal analysis have been of great utility to characterize the powders. These tests are done at the Divisão de Química do Instituto de Pesquisas Tecnológicas

## OBTAINED RESULTS

Table I presents the values of obtained results for the surface areas of the three types of powders. These values indicate a satisfactory sinterability according to the results obtained by Carpenter<sup>(1)</sup> by one of the authors.<sup>(3)</sup>

**Table I**  
Surface Area Bulk Density and O/U Ratio for the  
Uranium Oxides (After Reduction)

Material	Surface Area ( $\text{m}^2 \text{g}^{-1}$ )	Bulk Density $\text{g cm}^{-3}$	O/U Ratio
UO <sub>2</sub> N	2.7	1.44	2.04
UO <sub>2</sub> S	2.5	1.51	2.06
UO <sub>2</sub> D	1.9	1.24	2.01

Table II shows the results of a typical sizing analysis of the powders as reduced and after milling during 1 and 5 hours in a cylindrical mill 130 mm in diameter placed on rollers that with speed of  $85 \text{ cm s}^{-1}$  with 12.5 mm diameter steel balls

**Table II**  
Milling Effect on the UO<sub>2</sub> D for Different  
Operations Times

Type s Screen	Weight Retained Percentage		
	0 h	1 h	5 h
20	70.1	0.4	0.2
65	13.6	52.8	84.9
100	5.3	28.1	2.8
150	6.0	12.7	6.3
200	3.7	3.9	2.9
200	1.3	2.1	1.9

Both the type N and the type S UO<sub>2</sub> were easily classified under -20 mesh. Type D powder however after the reduction presented lumps of about 40 mm. Milling of this material in rubber lined cylindrical mills during 6 hours did not permit to improve substantially its grain size classification destroying only the large agglomerates. Therefore in the case of type D UO<sub>2</sub> it was necessary to modify the milling conditions.

Different samples of domestic uranium salts besides the ones of UO<sub>2</sub> produced by the authors at the Divisão de Metalurgia Nuclear were taken to sinterability tests in hydrogen at Argonne National Laboratory. It was then shown that the pellets produced met the current U.S. specifications for powder sinterability. It was necessary consequently to establish later on the know how for the sintering UO<sub>2</sub> in hydrogen atmosphere in order to get high density pellets. It is convenient to emphasize that this is the utilized process in all industrial plants for the production of millions of pellets which are being utilized in the nuclear power reactors.

The morphology of the powders of UO<sub>2</sub>-D as reduced is being studied through scanning electron microscope. The utilized microscope (figure 2) recently installed at the Divisão de Metalurgia

Nuclear of the IEA (september 1971) has permitted also some morphologic studies of preliminar character for the  $UO_2$ , N and  $UO_2$ , S oxides. Initially examinations were done in the ceramic particles without giving any metallic coating at low tensions as 1 to 2 kV later tensions up to 5 kV were used without development of excessive loads at the surface of the samples through the utilization of anti static fluid (Testenal Photowerk Hamburg). At present some sample are being prepared to receive a Au Pd film which enables one to use tensions up to 30 kV.



Figure 2 – Scanning electron microscope made by Cambridge Instrument Co. model S-4 installed at Divisão de Metalurgia nuclear.

The preliminary examinations were done with 2000 X for powders of -325 mesh. They presented the  $UO_2$ , D as compact agglomerates formed by round grains with small amount of pores. The  $UO_2$ , S presented also too dense agglomerates but their constituent grains were large and surrounded by a lot of large pores. The  $UO_2$ , N showed also some agglomerates but presents smaller grains and less pores than in the two previous cases.

Under sinterability standpoint, it is important to know the morphologic of the particles forming the agglomerates. For this objective it is important to coat the powders with metallic films which permits to reach tensions of the order of 20 kV with the possibility of resolutions better than 200Å.

### PRESSING AND PRE-SINTERING

For the sintering tests of the produced powders, compacts of 10 mm diameter were produced in a double effect die. The applied load were correspondent to the pressures of 1 and 1.5 ton/cm<sup>2</sup>. The authors tried to obtain the largest values of density without the existence of surface flaws.

The powders of types S and N needed frequently a previous operation of pre pressing at higher pressures between 1.5 and 2.5 ton/cm<sup>2</sup>. Only through this way it was possible to get pellets for such powders without surface defects and with densities larger than 4.5 g/cm<sup>3</sup>.

In order to remove progressively the agglomerate, it was used a pre sintering furnace in a nitrogen atmosphere. This furnace has eight zones of independent control of temperature and the operation reached 500°C at the hottest zone where the pellets stayed at this temperature for 60 minutes.

Table III shows the densities data obtained.



Table III

UO<sub>2</sub> Pellets Density (After Pre-Sintering  $d_{ps}$ )

UO <sub>2</sub> powders	$d_{ps}$ g cm <sup>-3</sup>	$d_t$ %
N	5.15	47.0
S	5.01	45.6
D	4.55	41.5

### SINTERING OF THE PELLETS

Two furnaces were utilized for the development of the sintering operations in a hydrogen atmosphere. The first used (figure 3) had a cylindrical muffle of alumina surrounded by a molybdenum wound electrical resistance. Its hottest zone reached 1670°C for 60 cm in length. The second furnace of Tammann type (figure 4) had a graphite muffle reaching 1600°C for 80 cm in length in this case the pellets were surrounded by powdered alumina and placed in graphite boats.

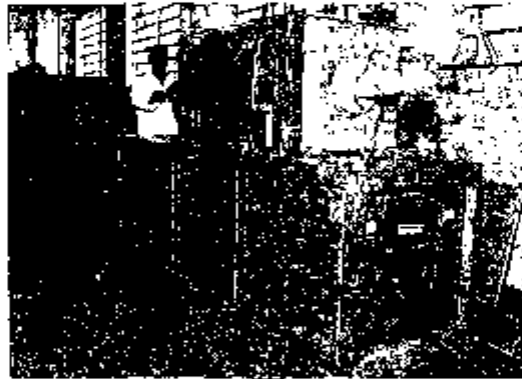


Figure 3 — Hedin furnace used for the sintering test at 1625°C with hydrogen atmosphere. This furnace use a molybdenum resistor for heating.



Figure 4 — Tammann type furnace used for sintering tests at 1625°C under hydrogen. The muffle is surrounded by the graphite tubular resistor.

It was noted that after 1200°C the UO<sub>2</sub> pellets suffered a severe attack in the Tammann type furnace. At 1600°C there was some desaggregation of the external layer of the pellets up to 2 mm deep. This occurrence, being studied now by the authors, could be explained due to the formation of uranium carbides on the sintered pellets and its later interaction with the hydrogen and the humidity existing at the furnace atmosphere.

It is to be noted the graphite resistors of Tammann furnaces is very much affected by the temperature. The cumulative experience in utilizing these furnaces shows that the resistors life increase from six hours at 1600°C to some ten hours at 1400°C. For these reasons, this type of furnace is used only for operations requiring temperatures less than 1400°C.

On the other hand, the alumina muffle furnace heated by molybdenum resistors has shown a very efficient operation but its W-WRe thermocouple only lasts about 2 hours when their tips are exposed to the furnace atmosphere. The temperature control has been made through an optical pyrometer and the W-WRe thermocouples are used only for calibrations of the optical pyrometer.

The hydrogen flow in this furnace is used to be about 12 l/min which is equivalent to about 0.2 l/min cm<sup>2</sup> of transversal section muffle area.

It has been observed that flows less than 0.2 l/min cm<sup>2</sup> caused partial oxidation of the UO<sub>2</sub> N.

The densities of the pellets sintered in this furnace correspond to 94.3% of the theoretical density. This value satisfies the N 5 North American and Canadian specifications.

The pellets present a very good surface in the sintered state. Some surface defects are eliminated by a centerless grinding.

Although the UO<sub>2</sub>-S and UO<sub>2</sub>-D powders gave high density pellets, they should be used for the sub-critical assembly or for low power critical reactors. If these oxides, therefore, are submitted to a comminution operation and pre-compacting under 1.5 t/cm<sup>2</sup> pressure, they are likely to provide pellets with densities about 98% of theoretical densities.

Table IV presents UO<sub>2</sub> pellets densities sintered at 1625 ± 25°C for 4 hours.

Table IV

UO<sub>2</sub> Sintering Pellets Densities (d<sub>s</sub>)

UO <sub>2</sub> powders	d <sub>t</sub> g cm <sup>-3</sup>	d <sub>s</sub> %
N	10.33	94.3
S	10.17	92.8
D	10.08	92.0

A polished section study of UO<sub>2</sub> pellets was examined under a Leitz Orthoplan ceramicographic microscope. The microstructure observed is in agreement with the Canadian MET-67 specifications.

Some other observations were made by the scanning electron microscope Stereoscan S-4 (figure 2).

Figure 5 shows the external surface of a  $\text{UO}_2$  D pellet without polishing with 1455 times magnification. This pellet presented sintered density of  $10.25 \text{ g/cm}^3$  and was obtained with 20 others sintered at  $1625^\circ\text{C}$  for 4 hours in the Hedén furnace. The heating rate was about  $140^\circ\text{C h}^{-1}$  and the cooling rate was  $300^\circ\text{C h}^{-1}$ . The micrograph obtained by the scanning electron microscope shows a good sintering and no defects in grain boundary.



Figure 5 — Pellet surface without polishing sintered at  $1625^\circ\text{C}$  for 4 hours under hydrogen atmosphere in the Hedén furnace made by  $\text{UO}_2$  D powder (1455 X). The micrograph obtained by a scanning electron microscope shows the perfect sintering and the grain boundaries without flaws.

Figure 6 presents the surface of a fractured pellet obtained in the same conditions above specified. It was made with  $\text{UO}_2$  N powder. Observe the  $\text{UO}_2$  grain fineness.

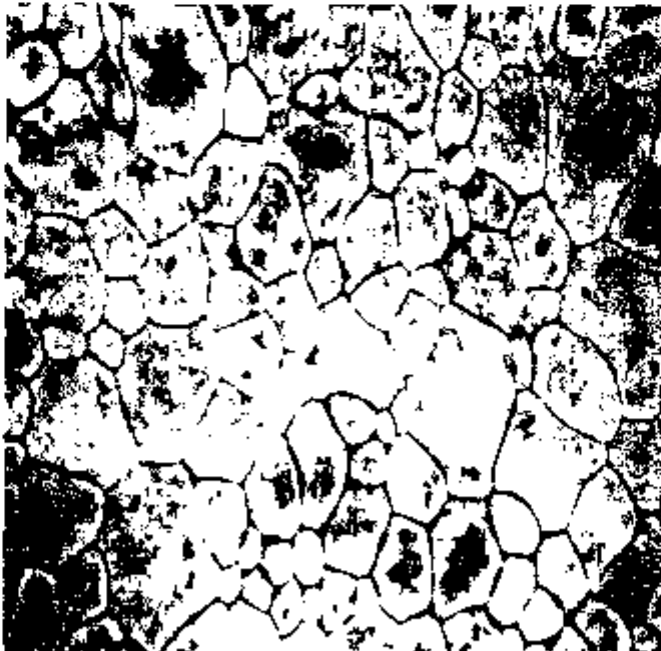


Figure 6 — Surface examination of a fractured pellet obtained in the same condition of that pellet of figure 5 made with  $\text{UO}_2$  N powder (1299 X). Note the fine grain texture of this  $\text{UO}_2$  pellet.

## PELLETS SURFACE GRINDING

The clearance between the cylindrical surface of pellets and the internal surface of the canning tubes is so small that surface grinding by a centerless machine is necessary

For the two projects being developed at the Divisão de Metalurgia Nuclear the following limits have been set

- a) maximum surface rugosity 2
- b) maximum diameter deviation 0.01 mm

There are also some other requirements on the surface finishing of the pellet basis

The sintered pellets are ground in three centerless grinders locally built

The values obtained are referred to a single centerless grinder although when in experimental production the three ones will be used

The grinding operation was made by vertical introduction because of the small number of test pellets. With a single pass the mean pellet diameter of  $8.30 \pm 0.06$  mm was reduced to  $8.202 \pm 0.006$  mm. These pellets were made by  $UO_2-N$  powder and have presented  $510 \text{ kg/cm}^2$  of Knoop hardness (Test with 100 g load). The mean grain size was 15.7 microns.

The grinding operation in present Pilot Plant operation will be carried out with three machines in series thus avoiding severe passes which would cause surface cracks. Experiences made in order to verify the influence of the severity of a pass on the surface cracks showed that reductions about 0.4 mm may cause these cracks which could be observed with a 5x magnifying glass.

The surface roughness of the ground pellets has been studied with scanning electron microscope making comparison between the electrical signals emitted from the ground surface and the ones emitted from a Knoop indentation. These first results showed that roughness is less than two microns. The figure 7 shows the pellet surface fractured by Knoop indenter penetration (1083x of magnification). This picture obtained by scanning electron microscope makes evident the intrinsic great brittleness of  $UO_2-S$  in comparison with the other two kinds of powders  $UO_2-D$  and  $UO_2-N$ .

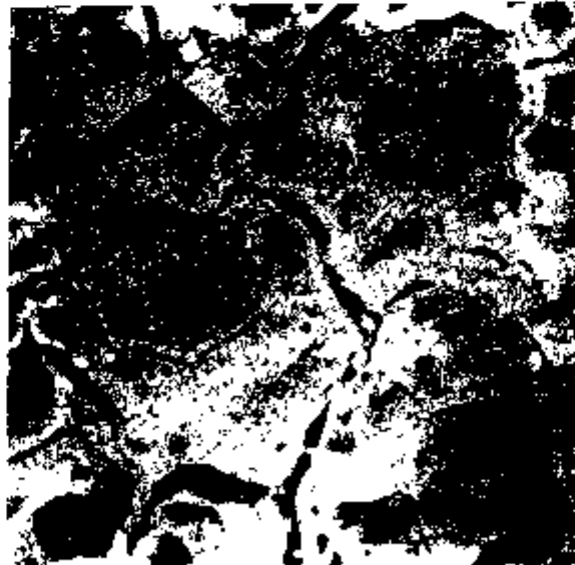


Figure 7 — Fractured pellet by Knoop indenter penetration (1083 X) obtained by scanning electron microscopy. This micrograph shows the  $UO_2-S$  inherent brittleness.

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- b) maximum diameter deviation 0.01 mm

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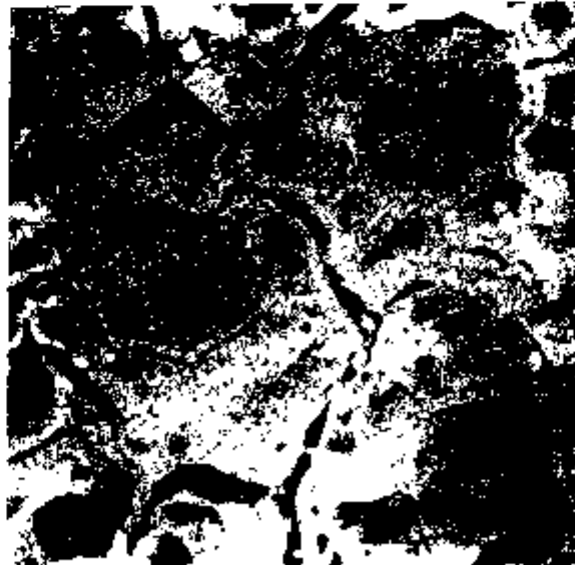


Figure 7 — Fractured pellet by Knoop indenter penetration (1083 X) obtained by scanning electron microscopy. This micrograph shows the  $UO_2-S$  inherent brittleness.

## CONCLUSIONS

1) The Divisão de Metalurgia Nuclear has been performed extensive studies on the influence of some  $UO_2$  powders properties on the manufacture of sintered  $UO_2$  natural pellets at  $1625^\circ C$  under hydrogen atmosphere.  $UO_2$  pellet fabrication within specified density, surface quality and dimensional reproductivity are the objectives of these studies.

2) The manufactured powders show wide differences in their characteristics as reflected by the values of particle density, particles average size, O/U rate, surface area and texture observed by optical microscopy and scanning electron microscope and in some cases by examination by X Ray diffraction and thermo differential analysis.

3) The sintering tests made with type N  $UO_2$  powders at  $1625^\circ C$  for 4 hours have given average densities of  $10.33 \text{ g/cm}^3$ . This value is equivalent to 94.3% of  $UO_2$  theoretical density. The pellets density manufactured under the same conditions with types S and D powders were 10.17 and  $10.08 \text{ g/cm}^3$  respectively which correspond to 92.8 and 92.0% of  $UO_2$  theoretical density.

4) The main results obtained by scanning electron microscope powder examinations were presented. These results showed large structural differences among those powders.

5) The pre-sintered pellets behaviour during sintering operations at  $1625^\circ C$  for 4 hours in hydrogen atmosphere proved that the type N  $UO_2$  powder was the best, due to the density values and surface quality obtained.

6) The ceramographic study of  $UO_2$  pellets was done with a Lertz Orthoplan microscope. The  $UO_2$  pellets satisfied the specifications used.

## RESUMO

Os autores descrevem o trabalho realizado na Divisão de Metalurgia Nuclear do Instituto de Energia Atômica na tecnologia de fabricação de elementos combustíveis.

Os resultados obtidos na sinterização de  $UO_2$  produzidos no próprio Instituto são discutidos com relação aos problemas encontrados na caracterização dos póis desse óxido. Os ensaios usados compreendem principalmente o de BET, o de Fisher e os de microscopia eletrônica de varredura e de transmissão.

Também são incluídos detalhes da Unia Pilota de Cerâmica Nuclear do Instituto de Energia Atômica relacionados com os programas a serem desenvolvidos em futuro próximo.

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