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A NEW FUNDAMENTAL HYDROGEN DEFECT IN ALKALI HALIDES!

Sporo Penha Morato² and Fritz Lüty³

ABSTRACT

Atom hydrogen in neutral (H⁰) and negative (H⁻¹) form on substitutional and interstitial lattice sites gives rise to well characterized model defects in alkali-holides (U, U₁, U₂, U₃ canters), which have been estandardly investigated in the part. When studying the photo-decomposition of OH⁻¹ defects, a new configuration of atomic charged hydrogen was dicovered, which can be produced in targe quantities in the crystal and is apparently not connected to any other impurity. This new hydrogen defect does not show any pronounced electronic absorption, but deplays a single sharp local mode band (at 1114 cm⁻¹ in KCI) with a perfect $\sqrt{2}$ H=0 instops shift. The defect can be produced by various UV or X-ray techniques in crystals doped with OH⁻, SH⁻¹ or H⁻¹ defects. A detailed study of its formation kinetics at low temperatures shows that it is primarily formed by the reaction of a mobile Cl₂ crownian (Hicenter) with hydrogen defects.

1 - INTRODUCTION

Hydrogen impurities alkali halide crystals are well studied entities in their configuration and properties. So far there are only four known basic forms of hydrogen centers in these fattices. Occupying one anion site we can either have an hydrogen ion (HT or U center as move commonly known^(1,2)) or a hydrogen atom (H^o or U₂ center⁽³⁾). In interstitual positions we can have again an hydrogen ion (H^{\circ} or U₁ center⁽⁴⁾) or an hydrogen atom (H^{\circ} or U₂ center⁽⁵⁾). All these forms of hydrogen centers are very accessible to a full study with optical and magnetic techniques that detect their electronic or vibrational transitions or even a possible paramagnetic character as shown by the Ua center for example. Hydrogen interstitials created by "optical radiation damage" through photodissociation of substitutional OH" or H", impurities have played and still play a model rule in the attempt to understand both the primary process and secondary reactions of the formed interstituals. In the case of the HT defect, an "extrincic changed Frenkel pair" consisting of anion vacancy and HT (instead of the CI1) is formed. The HT defect gives rise to a very broad structureless electronic absorption in the UV and to a local mode absorption in the IR (around 12µ for KCI). This latter displays at lowest temperatures, a fine structure consisting of several groups of sharp lines that correspond to H^{*}_i with various special correlations and interactions to the anion vacancy, or "correlated extrinsic Frenkel pairs" (the H₁ local modes are split by the interactions with the anion-vacancy). The thermal stability of the H₁ center shows the same behaviour of the intrinsic chlorine interstitials: thermal annealing curves with characteristic steps are observed. A first sharp annealing step corresponds to the extintion of the spaceally correlated extrinsic Frenkel pairs. A higher temperature annealing step corresponds to the extinction of the free uncorrelated H," defect.

Several "perturbed hydrogen centers" with the hydrogen located close to other impurities have been studied recently. Specially correlated U-H center pairs were found to display a broadening and shifting of the U center UV absorption and a removal of the degeneracy of its IR local mode absorption⁴⁸¹, Pairs of mutually perturbed U centers (H'H' peirs) were also the subject of recent

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investigations⁽⁷⁾. These < 110 > oriented H⁺H⁺ pairs produce several lines in the IR local mode region that could be interpreted with coupled oscillator models. The presence of additive mettalic impurities can also produce other forms of particled hydrogen as the U_A (Ne^o) center⁽⁰⁾. For example, a metallic impurities financity like Ne^o in KC) situated in one of the six nearest neighbor sites of an H⁺ center (then a U_A center) will cause the splitting of the triply degenerate local mode vibration of the H⁺ center. This splitting is due to the reduction of the O_A symmetry of the perfect lattice into C_{AN} symmetry.

2 - EXPERIMENTAL PROCEDURES

The KCI single crystal preparation and the experimental optical techniques have already bees described in detail in the preceding work¹⁰. To additively color our samples we employed the method described by Van Dorn¹¹⁰ with some modifications. In this method the F center concentration was obtained by a controlled vapor pressure of the metal vapor using an excess pressure of nitrogen gas and varied the liquid potassium temperature, thus varying its vapor pressure. Using an intermediate calibration curve from Van Dorn and Rogener's methods¹¹¹, which gives the F center concentration as a function of the potassium vapor pressure, we were able to obtain quite reproductible F center concentrations on the order of 10¹⁶ to 10¹⁸ cm⁻³. Hydrogenstion of an additively colored sample consists in the controlled diffusion of hydrogen into the crystal under high temperature and hydrogen pressure. By heating the crystal at 650°C under 20 atmospheres of hydrogen gas, hydrogen molecules diffuse into the crystal and react with the F centers, producing *sist*titutional hydrogen ions (U centers).

To irradiate our samples with X-Rays at low temperatures the cryosted head was provided with a fifth window made of a 1 mm thick aluminium foil at 45° bisecting the perpendicular optical paths. The X-Ray source was a Horelco with a W target. The unfiltered exit window placed at approximately 5 cm of the sample. The X-Ray exposure was always at 30 KV and 20 mA.

To investigate all the transparent range of the crystal with UV, V is and IR spectroscopy – especially in the hydrogen center local mode region we set up two experiments with the two parallel and perpendicular geometries using crystals of suitable dimensions. The aim was to produce the same photochemical reactions in both samples in parallel so we could carefully detect and correlate any IR effects to the already known spectral behavior in the UV region.

3 - RESULTS AND DISCUSSIONS

As it was shown in the previous work⁽⁸⁾, a sequence of UV irradiations at LNT in KCI:0H⁻ system sugarsted that a missing C_{1}^{0} crowdion was being trapped at some unknown form of hydrogen defect undetectable by UV or visible spectroscopy. With the changes in geometry previously described we set up an experiment that would show all the effects of this UV irradiation sequence at LNT by monitoring all the changes in the optical absorption of the KCI band gap.

2.1 - Photoproductions of a New Hydrogen Center

As shown in figure 1A we started with the photodissociation of the OH⁻ defect by intadiating monochromatic UV light at 204 nm (OH⁻ absorption band). As a result we observed in the UV range the OH⁻ \rightarrow H⁰ conversion as previously described. In the 1R range no significant spectral changes can be observed at this stage, as expected. The OH⁻ stretching vibration absorption at 2.7 μ is too weak to be detectable for the low OH⁻ doping used, and the neutral H⁰ interstitial does not give rise to an optical active local mode. In the next step we photo-destroyed H⁰ conters, reproducing our former UV spectral results. In this irradiation stage, new pronounced effects developed in the 1R range. Besides the appearance of the H⁺ local mode transitions — as expected from the UV result — we observed the growth of a new single band so far not reported in the literature. This band has its maximum at

1112 cm⁻¹ (at LNT), at a 30% higher energy compared to the H^{*}₁ center local mode transition. We call the center responsible for this new hydrogen local mode absorption tentatively H^{*}_x, with the "x" indicating its unknown structure and site, and the minus sign indicating that this new center should be charged by being IR active.

Proceeding with the series of spectral irradiations showed in Figure 1, we illuminated in the third stage with U-band light, producing the well known $H^{-+} H_1^-$ conversion. This step basically did not change the optical density of this new center. As expected we observed in both ranges (IR and UV) the decrease of the H⁻ center (U band) and basides the increase of the H₁⁻ center, the U band destruction is also responsible for some H_2^0/F center pairs formation. The new H₁⁻ defect apparently does not participe in all these transformations. A further U₃ band irradiation was performed in case B, mostly to follow in the IR the behavior of the new H₁⁻ band. The major effects after this irradiation were the increase of the H⁻ center, the increase of the H₁⁻ center due to the fact that U₃ band irradiation also reaches the U₁ electronic absorption band which is very broad and extends from 200 to 230 nm in the far UV. We should paint out here that in a similar set of experiments with a KCI:OH⁻⁺ Na⁺ crystel, the monochromatic U₃ band irradiation only brought up H_A⁻ centers as mentioned before with no traces of H₁⁻ center formation whatsoever in the IR region.

All the above described effects can happen simultaneously if we irradiate the crystal with und-spersed UV light. This was indeed observed by irradiating two KCI samples of different OHconcentrations. For the low OHT concentration sample we observed a parallel rise of the H_, H_ and HT center transitions (Figure 2A). After a prolonged irradiation these transition bands reached some sort of dynamical equilibrium and saturated. In several attempts we disturbed this dynamical equilibrium by illumination with monochromatic light into the U band for example, producing the $H^- + H^0_0 + \Box$ conversion already described in Figure 28. Conversely we would invert this process by ilumminating into the U₁ band. While the relative amounts of H^* and H_1^* local mode absorptions changed reversibly in these tests, the height of the H_ transition band remained approximately constant (as it did in the experiment) shown in Figure 18). At the end we were able to return to the dynamical equilibrium relative heights of these three transitions as shown in Figure 1A just by repeating for a few hours the full Xenon lamp intadiation. The IR local mode transitions and the corresponding UV spectral absorptions after ten hours of undispersed light irradiations are shown respectively in Figures 3A and 38. From the UV spectrum we can confirm that to new bands were found in the UV region. This is consistent with the fact that the Hg, apparently laving no significant electronic transition in the transparent crystal range, will be the most prominent product of the OH"photodecomposition at LNT, as observed from this saturation experiment.

We repeated this full Xanon kern kradiation experiment for a KCI:OH⁺+Na² sample (see Figure 28) and observed the same saturation behavior as the one obtained with a KCI:OH⁻ sample (figure 2A). Since our irradiation also contains the wavelength which is absorved by the H_A centers, this kind of undispersed irradiation provents their formation. (We confirmed the photodescruction of H_A centers in another experiment where we bleached the H_A band by irradiating monochrometic light that is absorbed by the H_A center). As a result of this saturation experiment we again ended up with H⁺_H centers as the most prominent band. For a higher conventionate KCI:OH⁻ sample, under the same irradiation conditions, we were able to see in some more detail the initial build-up of these three transition bands (Figure 4). The initial formation rate of the H⁺_A centers is law, but increases merkadly under prolonged irradiation. This means that under undispersed irradiation the H⁺_A transition band is formed at the expense of another reaction product which must be initially created by the UV light irradiation component.

3.2 - H" Loud Mode Speets and Leetops Effect H" Fermation in KBr:OH" and RbBr:OH" Crystals

A closer look at the H^{*}₂ transitions band revealed very interesting features. It did not show any structure or splitting when cooled to low temperatures, displaying a single band with maximum at

1114 cm⁻¹ and a halfwidth of 2.5 cm⁻¹ at 6K. The variation of the band shape as a function of temperature can be seen in Figure 5. Our general observations of the H^{*}_a center so far strongly indicate the presence of a hydrogen. But how would one confirm the presence of a hydrogen in this new defect? A rather straighforward test was to repeat the full Xenon lamp irradiation for a system such as KCI:OD⁻ and search for isotope shifts in the observed transitions. This experiment is described in Figure 6. We indeed observed the displacement of the H^{*}_a transition from 1112 to 779 cm⁻¹, an almost perfect $\sqrt{2}$ isotope shift (H + D). Parallel to this we also observed the isotope shift for the local mode of the '4⁻_a center from 845 cm⁻¹ to 600 cm⁻³. Finally, in a brief attempt to demonstrate the generality of the H^{*}_a defect and its photoproduction from OH⁻ defects, we took KBr:OH⁻ and RbBr:OH⁻ samples and repeated the full Xenon lamp irradiation experiment. We indeed found in these two systems the formation of H^{*}_a and H⁻_a centers as in the KCI:OH⁻ system. For KBr:OH⁻ the peak positions of H⁻_a and 745 cm⁻¹ and 790 cm⁻¹ respectively. For RbBr:OH⁻ they were at 1021 cm⁻¹ and 745 cm⁻¹ respectively. These results indicate the general nature of the H^{*}_a center excluding the possibility if its being formed only in KCI:OH⁻ systems.

Up to this point we know or anticipate that hydrogen and mobile Cl_{1}^{0} crowdions are the basic candidates for the formation of the H_{1}^{-} hand. However we must remember that so far we only dealt with one of the two products of the OH"photodissociation, namely the H_{1}^{0} center. In other words, the O" center transition into consideration although it was always present and in the same quantities as the initial H_{1}^{0} center. In the full irradiation experiments, the presence of the O"center is even more critical since here we are also irradiating in the O"band. This could result in photochemical reaction of new defect appreations. Since the precise optical detection of the O"center and this contribution of the various defect reactions was experimentally impractical, its exclusion from the formation of the H_{m}^{*} center can only be proven by indirect experiments.

3.3 - The Formation of H^{*} Centers in KCI:H^{*}

To answer all the pending questions that remained open up to this point we X-irradiated a KCI-H⁻ system at 77. The reasons to this are, a) X-rays are well known to produce Cl_{μ}^{0} crowdions in KCI it low temperatures^(1,2); b) a KCI-H⁻ system would offer hydrogen traps for the mobile crowdions; c) a KCI-H⁻ system would exclude the presence of the O⁻ centers. We X-irradiated a KCI sample containing 5.6 x 10⁻⁵ H⁻. Due to the high concentration of H⁻ defects, its local mode transition was monitored by the anti-Stokes phonon side band, since the main local mode transition is completely off scale (by a factor of ~ s0 higher than the phonon side band at ENT^(1,3)). After a prolonged X-irradiation we observe the formation of structures correspond to transitions of H⁺ centers that have different spatial correlations to their anti-centers — the anion vacancies ~ as explained by Fritz⁽⁴⁾. The X-irradiation also produces H⁺_a centers as we had antecipated (Figure 7A).

In Figure 8A we display the growth and decay curves of the centers involved in Figure 7A. We observe that X irradiation produces the following effects: a) H^* centers undergo a gradual reduction, reflected by the decrease of its phonon sideband; b) H^*_j centers measured by the main peak from the spatially uncorrelated ones are formed with approximately constant rate; c) H^*_j centers are formed initially with a small rate which increases under further X-irradiation. From C we recognize that H^*_{ij} centers are formed more efficiently with the help of some reaction product that is obtained during the initial X-irradiation exposure. This observation makes the H^*_j center the strongest candidate for a trapping site for the Cf⁰ crowdions.

In order to further verify the participation of H^{*} defects in the H^{*}₀ production process, we set up another experiment in which pre-exposed a KCI:H^{*} sample to UV irradiation. As a result of this UV irradiation we obtained, very efficiently, the H^{*} + H^{*}₁ conversion process without any trace of H^{*}₀ center formation (see Figure 78). Under X-irradiation of this previously UV exposed sample, we observed an initial decrease of H^{*}₁ centers and an increase of H^{*}₀ centers with an initial formation rate a factor of ~ 16 larger than the initial H^{*}₀ formation rate obtained by a direct comparison of Figure 20A with Figure 20B. Order further X-ray exposure the system tends to reach some dynamical equilibrium. observed facts confirm again strongly the pronounced participation of the H_i^- center as a trapping site for the CI_j^0 crowdion. We cannot decide from these experiments if H_i^- defects play any role as the CI_j^0 crowdion traps. We can however, conclude that if they participe in the formation process, this trapping probability for CI_j^0 crowdion is considerably lower than that of H_i^- defects.

A doubt that emerges after doing these X-ray experiments may be expressed by the following question: Could the KC!:H^{*} sample contain unwanted OH^{*} impurities which would have produced the H_{π}^{*} effects under X-irradiation? Due to the high concentration of the H^{*} center, their UV absorption would completely mask any small OH^{*} band. Two conclusive arguments can be given against such a possibility:

- a) The same UV light exposure which formed no H^{*}_x centers in the experiment of Figure 78 produced, in experiment shown in Figure 1. A an optical density of 0.15 in the H^{*}_x local mode absorption. If our KCI:H^{*} sample of experiment Figure 78 would have a factor of six lower concentration of OH^{*} impurity than experiment Figure 1A, we should have been able to produce 0.02 O.D. of H^{*}_x local mode absorption (0.02 O.D. was the lower limit of our detection system ability at the signal-to-noise ratio used). We can therefore say that W OH^{*} was present in the KCI:H^{*} sample it should be in concentrations less than 4 x 10⁻⁶;
- b) In another experiment we exposed a sample of low OH⁺ concentration (2.5 x 10⁻³) to X-rays at LNT during 16 hours and were unable to find any trace of H⁺_x, H⁺_y or H⁺ centers. This shows that X irradiation at LNT does not decompose at all the OH⁺ defact into any of the hydrogen reaction products H⁺_y, H⁺_y and H⁺_x as observed under UV irradiation of OH⁺. The participation of any unwanted OH⁺ additions in the experiments of Figures 19 and 20 is therefore definitely excluded, and so is the possible participation of any oxygen defects in the H⁺_y formation.

For a further confirmation of the non-participation of the O⁺ center in the H⁺_a formation we did another experiment in which we used a KCISH⁺ crystal and repeated the full UV irradiation treatment as we did with the KCIOH⁺ crystal. We know from previous work^{114,151} that the SH⁺ center can be also decomposed into S⁺ and H⁰_a defects. Proceeding with this full irradiation treatment we observed exactly the same H⁺_a band in the KCISH⁺ crystal. The experiments described in this section, besides confirming, the participation of H⁺_a and Cl⁰_a centers into the H⁺_a center formation, ruled out completely and possibility of the participation of the O⁺ center in the H⁺_a center production.

3.4 - Preliminary Conclusions About H^{*}₂ Formation from LNT Experiments

Up to this point the various experiments at LNT allow us to draw several conclusions about the nature and formation process of the new H_{\perp}^{*} center: a) under full UV light irratiation, H_{\perp}^{*} centers are the most prominent hydrogen reaction products of the OH^{*} photodecomposition; b) under stepwise monochromatic OH^{*} photodecomposition, H_{\perp}^{*} defects from in the secondary stage by photoexcitation of M_{\perp}^{0} centers. As this process creates mobile Cl_{\perp}^{0} crowdions, the latter are very likely candidates for the H_{\perp}^{*} formation; c) The 1R absorption of the H_{\perp}^{*} centers shows, by the $H \rightarrow D$ incorps shift, that it is caused by a perfectly localized vibration of a charged hydrogen ion; its single band structure indicates a site of high symmetry for the hydrogen, which does not split its local mode; d) H_{\perp}^{*} centers can be formed by X-irradiation of KCl:H^{*} crystals at LNT. This excludes any contribution of the oxygen in the H_{\perp}^{*} formation process, and confirms the idee that H_{\perp}^{*} centers are formed by the reaction of mobile Cl_{\perp}^{0} crowdions with hydrogen defects. The increase of the initial H_{\perp}^{*} formation rate in this experiment by a previous $H^{*} \rightarrow H_{\perp}^{*}$ conters in the interstitial H_{\perp}^{*} defect the most likely candidate for the trapping of the Cl_{\perp}^{0} crowdion forming the H_{\perp}^{*} center.

All the preceding experiments were done at LNT, where Cl_{j}^{0} crowdions are mobile right after their creation and thus form the H_{j}^{*} defects instantaneously. If this picture is correct, we should be able to produce H_{j}^{*} centers in controlled steps at lower temperatures where Cl_{j}^{0} crowdions are thermally stable. Experiments in this temperature range should therefore provide a

conclusive test on the formation process and a definitive identification of the hydrogen defect trapping the CI⁰ crowtion.

3.5 - Controlled Production of H_ Center at LHeT

We initially made the photodissociation of the OH* center at 77K (for experimental convenience) by monochromatic 204 nm light (Step 0 in Figure 9A). We then proceeded with the further photochemical reactions at 6 K. After photoclecomposing H^C₁ centers (Step 1, Figure 9A) at 8K we observed that the HT center local mode was built up as expected from previous experiments. In contrast to the corresponding experiment at LNT (see Figure 1), we observe at this step no trace of H land H.) formation. This confirms our previous assumption that the photodestroyed H⁰ centers are quantitatively converted into H° centers and Cl° crowdions, with the latter stabilized as H centers in the lattice. It further confirms our assumption thet only mobile Clo crowdions, reacting with hydrogen defects, are able to form the Hi centers. The "self-trapping" of the Cl⁰ crowdions as H centers therefore prevents the HT formation. If our line of arguing is right, the HT centers should be formed if we make the produced Cl⁰ crowdions mobile in some way. One way to achieve this is the optical excitation in the electronic transition (H band) of the Cl_{0}^{0} crowdion, which leads to an optically stimulated motion of the defect. If we shine monochromatic light into the it band (Step 2 in Figure 9A), we indeed observe the appearance of the H⁺₂ local mode band. Simultaneously we see the reduction of the H1 local mode band. As no H1 defects have been present during this reaction we must conclude that in this step H defects have been formed by the reaction of "optically mobilized" Cl_{i}^{0} crowdions with HT defects. A further irradiation into the Uz band (Step 3 in Figure 9B) proceeds with the H_{\perp}^{0} + H_{\perp}^{0} + Cl_{\perp}^{0} formation as in Step 1 without further H_{\perp}^{0} formation. A subsequent monochromatic irradiation in the U hand brought up mainly the local mode of the H7[] close pairs (Step 4, Figure 98) with the correspondent decrease of the H" absorption center. Now that we have two kinds of hydrogen centers competing as trapping sites, we again optically bleach the H band and observe further increase of the H_ local mode band. Simultaneously we see a considerable reduction in the local mode absorption of the H/() close pairs, and a very small reduction in the H⁻local mode. Apparently when both H⁻and H7(1) peris offered are much more effective for the H^{*}₂ formation. This is exactly what we observed in the X-ray experiment with the KCI.H" crystals of LNT.

The second possibility to mobilize the Cl_1^0 crowdions after their optical creation at LHeT is a thermal annealing process into the temperature range of their thermal instability (T > 55°). This process was performed as Step 6 in Figure 9D. It leads to a further considerable increase of the H_{\perp}^- centers and a simultaneous destruction of the $H_1^-/[.]$ close pairs. As the latter are clearly thermally stable in the used temperature range (they are thermally annealed only at T > 90K⁽⁴⁾), we conclude that the H_{\perp}^- increase was achieved at the expense of the $H_1^-/[.]$ close pairs. (The H⁻-centers remained approximately constant during the thermal annealing Step 6). Thus again we can conclude that thermally mobilized Cl_1^0 crowdions form H_{\perp}^- centers by reacting with $H_1^-/[.]$ close pairs. The involvement of the Cl_1^0 in the H_{\perp}^- formation is conclusively demonstrated in Figure 10, in which we monitor the absorption during the thermal annealing process. The correspondence of the thermal destruction of the Cl_1^0 centers (around 55 K) to the increase of the H_{\perp}^- absorption is clearly demonstrated (the change in the H-band absorption at 10K is caused by thermal reorientation of the Cl_1^0 crowdions).

A more general experiment confirmed the two above described results. We employed undispersed broad band UV irradiation at 6K as we previously did at 77K. The results of this experiment are displayed in Figures 11A and 118. From Figure 11A, Step 1, we see that after an exposure of undispersed UV light we observe immediate formation of all three hydrogen defects H_{π}^{*} , H_{μ}^{*} (correlated and uncorrelated) and H^{*} centers. Since our irradiation contains the wavelenghts of the OH^{*} band, of the different hydrogen centers involved IH, H_{μ}^{*} , H_{μ}^{0} , and of the H band absorption of the CI⁰₁ center, the simultaneous appearances of the H_{π}^{*} , H_{μ}^{*} and H^{*} centers is expected as in the LNT full irradiation cape

After this full UV exposite we bleached the H band using undispersed light but blocking only far UV light by proceed (b) ner with a cut off ther (see Step 2, Force 11A). As previously observed, we again see the raise of the H_{π}^{2} at the expense of H_{μ}^{2} centers. The following step was to practically repeat previous UV treatment to relestablish the dynamical equilibrium among the three defects (Step 3, Figure 11A) as previously done at LNT. As a last step, we performed again the thermal annealing to 77K procedure to make the crowdions thermally unstable. After this treatment we again observed an increase of H_{π}^{2} centers, a decrease of H_{μ}^{2} centers (both correlated and uncorrelated) and a corresponding decrease of H_{π}^{2} centers. This experiment, although being different in terms of irradiation procedures and intermediate products obtained, confirms the results of the previous monochromatic experiments.

3.6 -- The Thermal Destruction of the H_ Centers

To estimate the relative oscillator strength of the H_{π}^{*} center, we report to the previous work (Part I) where we learned that at LNT approximately 50% less H° centers are formed when compared to the LHeT experiment. We assume that the other 50% of "missing" HT centers are being consumed to form H⁺₂ centers. Using the relative strengths of the integrated absorptions in Figure 1B, we obtain $f_{H^{-}}/f_{H^{-}} = 0.7$. Considering the value 0.5 for the oscillator strength of the H⁺₂ center. After we have studied the detailed kinetics of the H^{*}₂ center creation and concluded that the H^{*}₂ center was the final and optically stable products of the $H_{\rm c}$ thermal desctruction. When heating a sample containing H centers and following their local mode absorption at LNT after pulse annealing to various higher temperatures we found that the H^{*} centers decay thermally in the temperature range 180-210K (Figure 12). This annihiling behavior is very close to the $H^{-}_{i} \rightarrow H^{-}_{i}$ thermal decay process of H^{-}_{i} interstitials described by $Fritz^{(4)}$. We indeed observed, together with the extinction of the H_x centers, this thermal reaction H +(] + H by the H decrease and H increase, as shown in Figure 12A. A simple comparative analysis of the strengths of absorption changes in the three IR bands involved in this process shows that the extinction of H⁺₂ centers does not create H⁺ centers. Taking the loss in absorption strength in the H⁺ local mode (Δ OD > 0.8) and converting it into the corresponding gain in absorption strength for the HT center, we would expect an optical density increase of ~1.7 for the H*absorption. We instead observed a mere 0.3 increase in O.D. of the HT local mode, a fact that by itself excludes the possibility that the destruction of H⁺ centers will form H⁺ centers. The small increase of the H⁺ local mode is fully accounted for by the thermal destruction of the H/[] extrinsic Frenkel pairs, as seen by the destruction of the H^{*} local mode band. No new local mode absorption in the IR range and no new electronic absorption in the UV/Vis range is observed to develop after the thermal destruction of the H⁺₀ defects.

3.7 - Final Conclusions on the H^{*}_____ Formation Process and Structural Model

The large variety of experiments described in the previous sections offered a consistent picture about the two components which participate in the dynamic H_{π}^{-} center formation process. We proved in different wave that CI_{1}^{0} crowdions are participating in the H_{π}^{-} center formation: a) in the LNT range, where U_{2} band irradiation creates the H_{π}^{-} defects, we showed that competing Na⁺ defects can capture and stabilize CI_{1}^{0} crowdions and suppress the H_{π}^{-} formation; b) creating CI_{1}^{0} crowdions by X-rays it LNT in a KCI H⁻ system leads to the formation of H_{π}^{-} (enters stepwise in a controlled way by first creating stable CI_{1}^{0} crowdions. Only when these crowdions were made mobile by optical or thermal excitation, the H_{π}^{-} defects appear.

The experiments under various temperature, irradiation, and defect conditions showed consistently the participation of hydrogen defects in the H_{π}^{-} formation process; a) in a crystal containing both $H_{\pi}^{+}(\Box)$ pairs and H^{-} defects, the formation of H_{π}^{-} centers is accompanied mostly by a reduction in close $H_{\pi}^{+}(\Box)$ pairs and very little, if eny, by a H^{-} defect reduction; b) If only H^{-} defects are present, the H_{π}^{-} formation proceeds at the expense of the H^{-} defects. We therefore conclude that the H_{π}^{-} defect is formed by the reaction of mobile Cl⁰ crowdions with either close $H_{\pi}^{+}(\Box)$ pairs (preferred process), or with H^{-} defects. As the $H_{\pi}^{+}(\Box)$ Frenkel pairs is = in terms of its net structural components - equivalent to the substitutional H defect, both these hydrogen traps for the Cl⁰ crowdion can lead to the same exist product. From the H_{π}^{-} local mode strength, spectral shape and isotope shift we concluded that the

defect must consist of a charged localized hydrogen defect in a site of high symmetry which does not split the local mode. The high frequency of the H_{μ}^{*} local mode indicates a stronger vibrational potential of the H_{μ}^{*} compared to the H_{μ}^{*} defect. From the thermal destruction of the H_{μ}^{*} canter we know that it is not converted back into H_{μ}^{*} centers, but disappears into some unknown optically inaccessible structure.

A structural model must be able to integrate and satisfy all these static and dynamic experimental features. We see only one possibility to achieve this in a single model: A hydrogen ion H⁻ in a body-centered interstitial position, with a trapped hole shared symmetrically by the fear surrounding nearest neighbor anione (see Figure 13). This model first satisfies the observed static features, a localized charged hydrogen defect in a site of high symmetry with a single unsplit three-fold degenerate local mode transition. A side from the trapped hole, it is similar to the H⁻₁ defect (see Figure 13). In the latter one, the charged interstitial H⁻₁ defect will have strong electrostatic interactions with the surrounding ions, repelling and shifting outwards, the anions, and attracting and pulling inwards the cations. The addition of a hole (positive charge) to the shell of nearest neighbor anions in the H⁺_x center will reduce the H⁻₁<> Cl⁻ repulsion and thus produce a closer distance between the hydrogen and the chlorine ions. As the potential for the localized vibration of the H⁻ is mainly produced by the Born Mayer repulsion interaction with the (large size) anions, we will expect a higher local mode frequency of the H⁺_x compared to the H⁺_y defect. This is in agreement with the frequency shift to higher energies found experimentally.

For the dynamic formation process, we have to regard the two cases: a) reaction of a CI a crowdion (an interstitial CI) ion with a bound hole) approaches a H7/D Frenkel pair, the most natural process to assume is the recombination of the Cl[®] interstitial with the empty racancy of the Frenkel pair. As a result we are left with the H7 interstitial and the hole which was carried previously by the CIP crowsion. If this hole just recombines with the interstitial Hi we would restore the original neutral interstitual H⁰ center. Apparently, however, the polarization of the surrounding ions around the H deffect. (Figure 13) make it possible to trap the hole at the CIT ions surrounding the HT defect in a stable way, b) reaction of a Cl^0 with a H⁺ defect in order to form the same H⁺_R defect in this reaction, the approaching CIT interstitial crowdion will exchange the lattice place with the HT defect, so that an interstitial H⁺ion is formed. The hole, carried by the Cl⁰ crowdion then gets bound to the H⁺ defect in the same way as above. In terms of its net components (after recombination of the hole with the Hi defect), the H_ is a neutral interstitial hydrogen atom. Thus when it gets thermally unstable and recombines with another H^{*}_____ detect, it would form a neutral interstitial H₃ molecule. This defect is known to be present in alkali halides and spectrally invisible in both UV and IR range. The "spectral dissppearance" of the H_ defect after thermal annealing is therefore will understandable with our model

We should point out that the proposed model is constructed strictly from an extended static and dynamic experimental material, which is conclusive and does not leave any alternative choice for a different model, consistent with the experiments. A theoretical justification for the proposed interstitiel H⁻ structure with a stabilized hole shared by the surrounding anions is beyond the scope of this work. The theoretical understanding and justification for this peculiar "inverted and self-trapped exciton" at an interstitial hydrogen defect appears to be a challenging and interesting problem for further theoretical studies.

RESUMO

Atomos de hidrogénio regetivos (H^{*}) ou neutros (H^{*}) quando em posições intersticiais ou substitucionais dela origem a deferios monteio para ceracterizados nos habitos abalinos (centros U, U₁, U₂, U₃) que tem sido extergivemente estudados no passado. Estudendo es a fotadecomposição de defeitos OH^{*} uma configuração nove de hidrogênio atômico cerregado foi encontrada Esta forma de hidrogênio pade ser producted em grandes quentidades no cristal e não está aperemente vincuiade com nenhuma outra impunsão. Este defeito de hidrogênio atômico cerregado nove de nenhuma abalición de pronunciade mais apresenta um único modo local de mésimo em 1114 cm⁻¹ em KCI e som um depiro porteco parteito de $\sqrt{2}$ (H + D). Este defeito pade ser productedo por váries técnicae UV ou de ratos X em cristes informativa com impuneram OH, SH^{*} ou H^{*}. Um estudo destinalita de sus cinítica de formação à balixae temperaturas mistra que este defeito é primariamente formação de sus cinética de númerio (Crowdion)^{*}. Cl^{*}₂ (centro II) dom deferiço de hidrogênico.



Figure 1 A -- Electronic absorption bands as a function of irradiation time for a sequence of monochromatic irradiations in a KCEOH crystal at LNT (thickness ~ 0.5 mm),

E. Local mode transitions in the IR for the same sequence of irradiations as in A (thickness - 8.0 mm)



Figure 2 — Plot of the local mode transition of H , H^{*}₁ and H^{*}₂ centers as a function of irradiation time under full Xenon lamp irradiation at 85K (measured at 77K).

- A KCI:OH" system.
- 8 KCI OH" + Na" system



Figure 3 A — Hydrogen centers local mode transitions after 10 hours of full Xenon lamp irradiation at 85K (measured at 77K).

8 - The respective UV spectra



Figure 4 - Plot of the local mode transitions of H⁺, H⁺_i and H⁺ centers as a function of irradiation time under full Xenon lamp irradiation for a KCI:OH⁺ c ystal with higher OH⁺ concentration.



Figure 8 – Temperature dependence of the spectral shape and halfwidth of the H_{π} local mode transition



Pipers 6 - Hotope shift effects of the H^{*}₂ and H^{*}₁ in a KCI:OD^{*}crystal after full Xenon lamp irradiation et 85K (measured at 77K).



Figure 7 A – Formation of H_{π}^{*} and H_{i}^{*} (of different spatial correlations) centers in a KCI + 5.6 x 10^{-5} H⁻ sample under X-rays irradiation at 77K.

 $B \rightarrow$ The same as above but after a previous UV irradiation, producing $H^* \neq H^*_1$ conversion.



Figure 5 A - Increase of H^{*}₀ and H^{*}₁ (uncorrelated ones) as a function of X irradiation time. The consumption of H^{*} centers is monitored by the phonon sideband of the H^{*} local mode.

B - The same as above but after the previous UV irradiation. All data are taken from the spectral measurements in Figure 19



Figure 9 — Local mode transitions of H", H" and H" under six different subsequent monochromatic irradiations at 6K (steps 0 = 5), and a final thermal annealing to 80K (step 6).



Figure 10 - H center decay and H^{*}₂ build up curves vs. temperature. This measurement corresponds to the experiment shown in Figure 21D.



- Figure 11 Production of hydrogen centers in KCI:OH" at 6K under undispersed light (broad band Interference filter + Xenon lamp).
 - A ~ Step 1. Formation of H^{*}, H^{*}_n (different correlations) and H^{*}_n centers under full UV irradiation. Step 2. H^{*}_n formation and H^{*}₁ destruction under H band (Cl^o₁ crowdion) light. Step 3. Repetition of Step 1.
 - H^{*}₁ production after a thermal annealing process to 77K, K^{*}₁ and H^{*} centers decrease after this treatment.



Figure 12 A = $H_{\pi^{+}}^{*}$, H_{1}^{*} and H^{*} centers local mode transitions before and after pulse annealing to 240K. B = $H_{\pi^{+}}^{*}$ center annealing curve at LNT taken under a pulse annealing procedure.



Figure 13 — Structural model of the H^{*}_{ii} center (A) and the H^{*}_i center (B) with indication of the shifts in position of the surrounding anions and cations due to electrostatic interaction with the interstitial hydrogen ion.

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