

Energy level decay processes in Ho^{3+} -doped tellurite glass relevant to the $3\ \mu\text{m}$ transition

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The primary excited state decay processes relating to the ${}^5\text{I}_6 \rightarrow {}^5\text{I}_7 \sim 2.9\ \mu\text{m}$ laser transition in singly Ho^{3+} -doped tellurite (TZBG) glass have been investigated in detail using time-resolved fluorescence spectroscopy. Selective laser excitation of the ${}^5\text{I}_6$ energy level at 1151 nm and ${}^5\text{I}_7$ energy level at 1958 nm has established that the rate of energy transfer up-conversion between holmium ions excited to the ${}^5\text{I}_7$ level is negligible for Ho^{3+} concentrations up to 4 mol. %. Excited state absorption was not observed from either the ${}^5\text{I}_7$ or ${}^5\text{I}_6$ levels and the luminescence from the ${}^5\text{I}_7$ and ${}^5\text{I}_6$ energy levels was measured to peak at $\sim 2050\ \text{nm}$ and $\sim 2930\ \text{nm}$, respectively. The ${}^5\text{I}_6$ level has a low luminescence efficiency of $\sim 8.9\%$ due to strong nonradiative multiphonon relaxation. In contrast, decay from the ${}^5\text{I}_7$ level is essentially fully radiative. A linear decrease in the decay time of the ${}^5\text{I}_6$ level with Ho^{3+} concentration augmentation results from energy transfer to OH^- ions in the glass (with $N_{\text{OH}} \sim 8.2 \times 10^{17}\ \text{ions cm}^{-3}$) and reduces the luminescence efficiency of the ${}^5\text{I}_6$ level to 8% for $[\text{Ho}^{3+}] = 4\ \text{mol. \%}$. Numerical simulation of a fiber laser incorporating 4 mol. % Ho^{3+} showed that a population inversion of $\sim 7.8\%$ is reached for square pulses of $100\ \mu\text{s}$ duration and a repetition frequency of 20 Hz at a moderate pump intensity of $418\ \text{kW cm}^{-2}$ if energy transfer to OH^- radicals is neglected. © 2011 American Institute of Physics. [doi:10.1063/1.3587476]

I. INTRODUCTION

Rare earth doped fiber lasers developed for emission in the infrared wavelength region longer than $2\ \mu\text{m}$ are of great interest for several applications, e.g., single frequency sources in this wavelength region would allow eye-safe coherent detection at long distance as a result of a number of atmospheric transmission windows that are available for $\lambda > 2\ \mu\text{m}$. The efficient detection of various molecules would also be possible because they have strong vibrational absorption lines in this wavelength region, i.e., in the so-called molecular “fingerprint” region.^{1,2} The use of the fiber geometry for the generation of shortwave and midwave infrared laser radiation introduces good thermal management and a comparatively low threshold because of the extended longitudinal dimension and small transverse cross section relevant to optical fibers.³

The ${}^5\text{I}_6 \rightarrow {}^5\text{I}_7$ and ${}^4\text{I}_{11/2} \rightarrow {}^4\text{I}_{13/2} \sim 3\ \mu\text{m}$ four-level transitions of Ho^{3+} and Er^{3+} offer broadband fluorescence and they have been successfully used to produce fiber lasers emitting in the shortwave infrared region in fluoride glasses.⁴⁻⁶ The Er^{3+} transition has received the most interest because fiber lasers based on this transition can be pumped with commercial diode lasers and an energy transfer up-conversion (ETU) process offers the potential for slope efficiencies beyond the Stokes limit. Pump excited state absorption (ESA) is strong, however, and may limit the maximum

achievable output power as a result of thermal and photo-induced absorption losses.

The Ho^{3+} ion, on the other hand, offers a longer free-running wavelength and little ESA, however, ETU cannot recycle the excitation and pump excitation is achieved with narrowly available diode lasers. Despite these limitations, the potential for high-power efficient operation of a $\sim 3\ \mu\text{m}$ fiber laser based on the Ho^{3+} ion is strong and it is useful to study the characteristics of the ${}^5\text{I}_6 \rightarrow {}^5\text{I}_7$ transition of Ho^{3+} when doped in robust materials that could allow for high-power pumping without the risk of optical or thermal damage. This investigation therefore involves a detailed spectroscopic study of Ho_2O_3 -doped tellurite (TZBG) glasses with varying Ho_2O_3 concentrations. Tellurite glasses have received a lot of interest because of their low maximum phonon energy (of $\sim 800\ \text{cm}^{-1}$) and robust thermomechanical properties. In the this study, we reveal the important radiative and nonradiative energy level decay processes that relate to the ${}^5\text{I}_6 \rightarrow {}^5\text{I}_7$ transition in tellurite glass after selective energy level excitation. Numerical simulations have been used to calculate the population inversion for the ${}^5\text{I}_6 \rightarrow {}^5\text{I}_7$ laser emission near $3\ \mu\text{m}$ in a Ho^{3+} -doped (4 mol. %) tellurite (TZBG) glass fiber laser for cw pumping at 1153 nm.

II. EXPERIMENTAL PROCEDURE

The Ho^{3+} -doped germanium tellurite (TZBG) glass samples used for the time-resolved luminescence spectroscopy were prepared from high-purity raw materials: 99.99% purity

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TeO₂, ZnF₂, and Bi₂O₃; and 99.999% purity GeO₂. The glass composition was $(100 - x) \times [74.5 \text{ TeO}_2 - 12.2 \text{ ZnF}_2 - 4.9 \text{ Bi}_2\text{O}_3 - 6.4 \text{ GeO}_2] + x \text{ Ho}_2\text{O}_3$ with $x = 0.5, 1, 2,$ and 4 mol. \% . The starting powder materials were melted in a Pt-Au crucible in a dry glovebox environment at 900 °C for 30 min to reduce the OH⁻ content in the glass. The molten liquids were poured into polished brass molds and annealed at 315 °C for 4 h to remove any mechanical stress. The glass density was 5.89 g cm⁻³ and the measured refractive index was 1.98 at 2750 nm. The samples were cut and polished into $4.7 \times 10 \times 15 \text{ mm}^3$ rectangular prisms. The Ho³⁺ density was calculated to be $1.07 \times 10^{20} \text{ ions cm}^{-3}, 2.14 \times 10^{20} \text{ ions cm}^{-3}, 4.27 \times 10^{20} \text{ ions cm}^{-3}$ and $8.46 \times 10^{20} \text{ ions cm}^{-3}$ for the tellurite samples having $x = 0.5, 1, 2,$ and 4 mol. \% , respectively.

The absorption spectra in the range of 2000–10 000 nm were measured using a FTIR spectrophotometer. The decay characteristics of the excited states of Ho³⁺ were measured using pulsed 10 mJ (4 ns) laser excitation from a tunable optical parametric oscillator (OPO) pumped by the second harmonic of a Q-switched Nd-YAG laser (Brilliant B from Quantel). Tunable laser excitation from the OPO was used to excite the ⁵I₆ energy level at 1151 nm and ⁴I₇ energy level at 1958 nm. The infrared luminescence (for $\lambda > 1000 \text{ nm}$) was detected using an InSb infrared detector (Judson model J-10 D cooled to 77 K) in conjunction with a fast preamplifier with a response time of $\sim 0.5 \mu\text{s}$, and analyzed using a digital 200 MHz oscilloscope (Tektronix TDS 410). The visible and near infrared (i.e., $\lambda < 1100 \text{ nm}$) was investigated using a photomultiplier tube (EMI) with a sensitive cathode of the S-1 or S-2 type (refrigerated to -20°C); the photomultiplier has a response time of 20 ns. All the fluorescence decay characteristics were measured at 300 K. To isolate the infrared luminescence signals, bandpass filters each with $\sim 80\%$ transmission at 1200 nm or 2750 nm with a half width of 25 nm and an extinction coefficient of $\sim 10^{-5}$ outside this band were used.

The optical absorption spectrum of Ho³⁺ ions in tellurite glass has two main features in the near infrared, one feature at around 1958 nm from the ⁵I₈ → ⁵I₇ transition and a second feature near 1151 nm due to ⁵I₈ → ⁵I₆ transition. When the Ho³⁺-doped material is excited at 1151 nm the following processes⁷ are known to occur in fluoride glass:

- (a) Ground state absorption (GSA); $\text{Ho}^{3+} (^5\text{I}_8) + h\nu (1151 \text{ nm}) \rightarrow \text{Ho}^{3+} (^5\text{I}_6)$
- (a') Excited state absorption (ESA); $\text{Ho}^{3+} (^5\text{I}_6) + h\nu (1151 \text{ nm}) \rightarrow \text{Ho}^{3+} (^5\text{S}_2)$
- (b) Energy transfer up-conversion (ETU1); $\text{Ho}^{3+} (^5\text{I}_7) + \text{Ho}^{3+} (^5\text{I}_7) \rightarrow \text{Ho}^{3+} (^5\text{I}_6) + \text{Ho}^{3+} (^5\text{I}_8)$
- (b') Energy transfer up-conversion (ETU2); $\text{Ho}^{3+} (^5\text{I}_6) + \text{Ho}^{3+} (^5\text{I}_6) \rightarrow \text{Ho}^{3+} (^5\text{F}_5) + \text{Ho}^{3+} (^5\text{I}_8)$
- (c) Cross-relaxation (CR); $\text{Ho}^{3+} (^5\text{S}_2) + \text{Ho}^{3+} (^5\text{I}_8) \rightarrow \text{Ho}^{3+} (^5\text{I}_4) + \text{Ho}^{3+} (^5\text{I}_7)$

Each of the processes listed above has been shown to affect the operation of the the 3 μm laser in the cw pumping regime in Ho³⁺-doped ZBLAN glass.⁷

III. EXPERIMENTAL RESULTS

Figure 1 shows the visible to shortwave (a) and mid-wave (b) absorption spectra of Ho³⁺ (2 mol. %)-doped tellurite

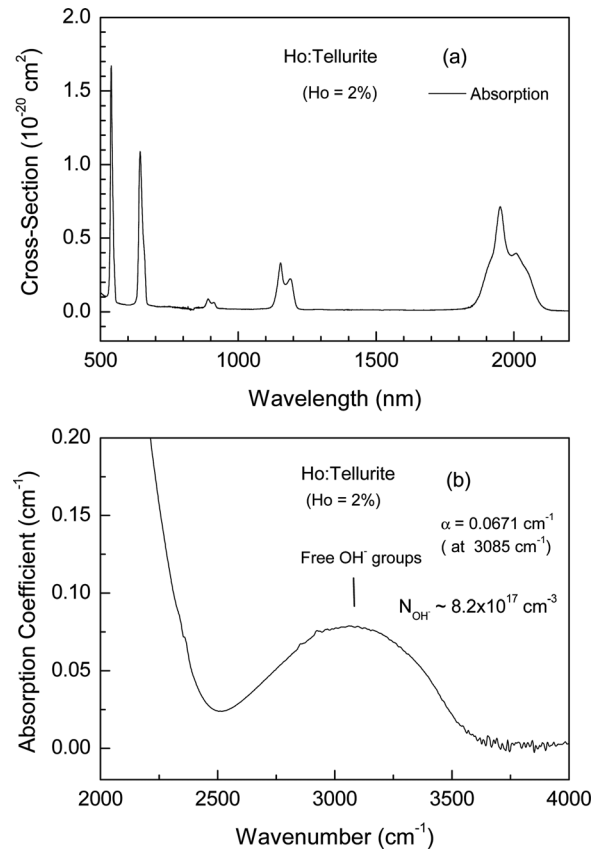


FIG. 1. Measured absorption transitions in the (a) visible and shortwave infrared region and (b) OH⁻ vibration feature in Ho³⁺ (2 mol. %)-doped TZBG glass (sample thickness = 4.7 mm).

ite glass. The spectrum shows a broad strong absorption band between 2500 cm⁻¹ and 3500 cm⁻¹, which we attribute to free OH⁻ groups.⁸ The narrow absorption band due to isolated OH⁻ radicals, with maximum absorption at $\sim 3735 \text{ cm}^{-1}$, was not observed. The concentration of OH⁻ radicals is given by $N_{\text{OH}} = \frac{N_{\text{Av}} \alpha}{\xi}$, where N_{Av} is Avogadro's constant (6.02×10^{23}), α is the absorption coefficient ($= 0.0672 \text{ cm}^{-1}$, relevant to the OH⁻ vibration band at 3085 cm⁻¹), and ξ is the absorptivity of free OH⁻ groups in the glass. Using $\xi = 49.1 \times 10^3 \text{ cm}^2 \text{ mol}^{-1}$ (Ref. 9), we estimated the OH⁻ concentration in our samples to be $8.24 \times 10^{17} \text{ ions cm}^{-3}$; a value ~ 29 times smaller than the free OH⁻ density (of $2.4 \times 10^{19} \text{ cm}^{-3}$) found in Er³⁺-doped germanotellurite glass produced from drying the melt using oxygen gas and CCl₄ bubbling.¹⁰

A. Infrared fluorescence spectrum of Ho³⁺ in tellurite (TZBG) glass

The fluorescence emission spectrum of Ho³⁺ in the shortwave infrared was measured for [Ho³⁺] = 4 mol. %. Figure 2 shows that there are two emission bands; one from the ⁵I₆ level at $\sim 2930 \text{ nm}$ and one from the ⁵I₇ level at $\sim 2050 \text{ nm}$. The emission cross section due to ⁵I₆ → ⁵I₇ and ⁵I₇ → ⁵I₈ transitions was calculated using $\sigma_{\text{emis}}(\lambda) = \frac{\bar{\lambda}^4}{8\pi n^2 c} A_{ij} \frac{S(\lambda)}{\int S(\lambda) d\lambda}$, where $\bar{\lambda}$ is the average emission wavelength or centroid, A_{ij}

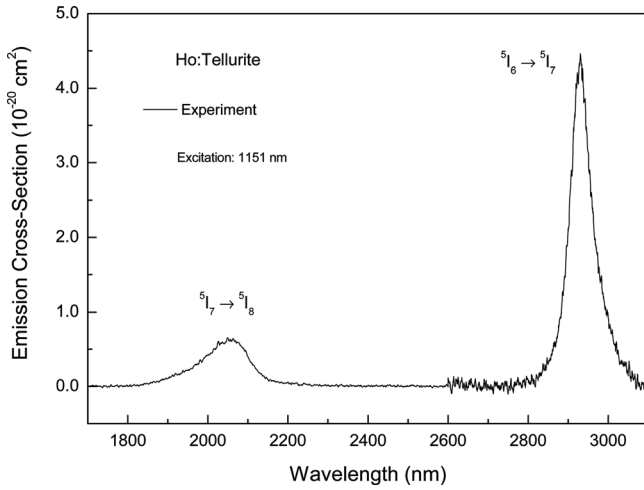


FIG. 2. Measured emission spectrum of Ho^{3+} (4 mol. %) -doped tellurite (TZBG) glass using a pulsed laser excitation at 1151 nm with an average energy of 13 mJ and pulse duration of 4 ns. Spectrum was measured using a boxcar technique.

(s^{-1}) is the radiative transition probability, $S(\lambda)$ is the line shape of the emission band, $\int S(\lambda)d\lambda$ is the integrated line shape, n is the refractive index ($= 1.98$), and c is the speed of light.

The emission cross section of the ${}^5I_6 \rightarrow {}^5I_7$ transition was calculated using the radiative transition probability $A_{ij} = 138 s^{-1}$,¹¹ the experimental value of $\bar{\lambda} = 2930$ nm, and $S(\bar{\lambda})/\int S(\lambda)d\lambda = 1.25 \times 10^{-2} \text{nm}^{-1}$, which gave an emission cross section of $4.30 \times 10^{-20} \text{cm}^2$ at the maximum emission wavelength of 2930 nm. The emission cross section of the ${}^5I_7 \rightarrow {}^5I_8$ transition was calculated using the radiative transition probability $A_{ij} = 1/\tau_R$ (where $\tau_R = 5.8$ ms calculated below), the experimental value of $\bar{\lambda} = 2050$ nm and $S(\bar{\lambda})/\int S(\lambda)d\lambda = 6.16 \times 10^{-3} \text{nm}^{-1}$, which provides an emission cross section of $6.37 \times 10^{-21} \text{cm}^2$ at 2050 nm. The radiative lifetime of the 5I_7 level was calculated according to the relationship¹² $\tau_{\text{rad}} = \frac{(2J+1)}{(2J'+1)} \frac{\bar{\lambda}^4}{8\pi c n^2 \int \sigma_{\text{abs}}(\lambda)d\lambda}$, where τ_{rad} is the radiative lifetime, J and J' the total momentum for the upper and lower levels, $\int \sigma_{\text{abs}}(\lambda)d\lambda$ is the integrated absorption cross section of the 2 μm band, and $\bar{\lambda}$ the mean wavelength of the absorption band. The radiative lifetime of the 5I_7 level was calculated using the $\bar{\lambda} = 1964$ nm and the integrated absorption cross section $\int \sigma_{\text{abs}}(\lambda)d\lambda = 7.7016 \times 10^{-26} \text{cm}^3$ obtained from the spectrum shown in Fig. 1(a), giving $\tau_{\text{rad}} = 5.8$ ms, which is longer than the 3.52 ms for this lifetime given in Ref. 11.

B. Emission decay from the 5I_6 level

Figure 3 shows the emission decay characteristic at 1200 nm for $[\text{Ho}^{3+}] = 0.5$ and 2 mol. % after laser excitation at 1151 nm with a mean pulse energy of 10 mJ and 4 ns pulse duration. The luminescence decay of 5I_6 excited state was nonexponential and was fitted using the Burshtein model,¹³

$$I(t) = I_0 \exp(-\gamma\sqrt{t} - t/\tau_m), \quad (1)$$

where γ ($s^{-1/2}$) is the transfer constant due to the direct donor to acceptor transfer and τ_m is defined by

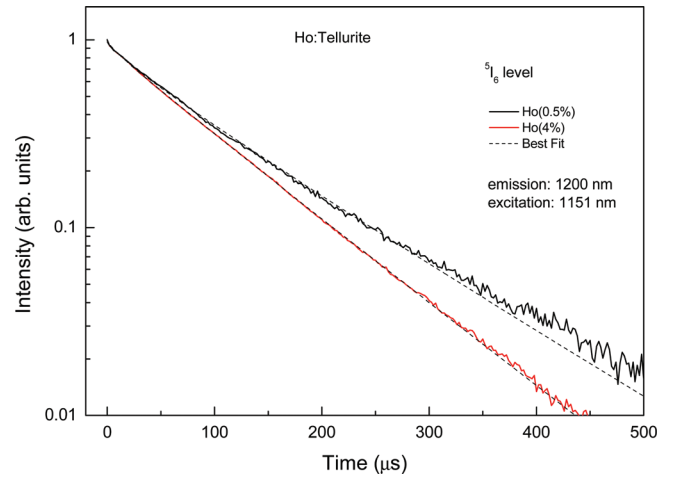


FIG. 3. (Color online) Measured emission decay characteristic (solid lines) of the 5I_6 level of Ho^{3+} (0.5 and 4 mol. %) -doped tellurite (TZBG) glass using short pulse laser excitation at 1151 nm with an average energy of 10 mJ and pulse duration of 4 ns at 10 Hz. Broken lines indicate the best fit using the Burshtein model of Eq. (1).

$$1/\tau_m = 1/\tau_R + W_{\text{nr}} + \omega, \quad (2)$$

where τ_R is the radiative lifetime and ω is the transfer constant (s^{-1}) due to the migration assisted donor to acceptor transfer. W_{nr} is the nonradiative multiphonon decay rate (s^{-1}). Because the luminescence decay of 5I_6 level at 1200 nm was nonexponential, the effective 5I_6 lifetime (τ) was obtained by integration,¹⁴ according to

$$\tau = \frac{1}{I_0} \int_0^\infty I(t) dt, \quad (3)$$

The best fit to the measured 1200 nm luminescence decay curve was carried out using Eq. (1); see dotted line in Fig. 3. The fitting parameters were: (i) $\gamma = 32.4 s^{-1/2}$, $\tau_m = 136.8 \mu\text{s}$, and $\tau = 99.2 \mu\text{s}$ for $[\text{Ho}^{3+}] = 0.5$ mol. % ($R^2 = 0.997$); (ii) $\gamma = 28 s^{-1/2}$, $\tau_m = 126 \mu\text{s}$, and $\tau = 96.3 \mu\text{s}$ for $[\text{Ho}^{3+}] = 1$ mol. % ($R^2 = 0.999$); (iii) $\gamma = 23.4 s^{-1/2}$, $\tau_m = 119.3 \mu\text{s}$, and $\tau = 95.7 \mu\text{s}$ for $[\text{Ho}^{3+}] = 2$ mol. % ($R^2 = 0.999$); and (iv) $\gamma = 14.8 s^{-1/2}$, $\tau_m = 102.1 \mu\text{s}$, and $\tau = 89.6 \mu\text{s}$ for $[\text{Ho}^{3+}] = 4$ mol. % ($R^2 = 0.999$).

Figure 4(a) shows that the decay time measured for 5I_6 level is linearly dependent on the Ho^{3+} concentration in tellurite (TZBG) glass. This effect was attributed to energy transfer from the 5I_6 level to OH^- radicals (i.e., free OH^- groups) that are present in the samples with the estimated concentration of 8.24×10^{17} ions cm^{-3} . The rate of energy transfer to the radicals was obtained using

$$W_t = 1/\tau - 1/\tau_R - W_{\text{nr}}. \quad (4)$$

The intrinsic lifetime (τ_d) of the 5I_6 level was obtained by extrapolating the line in Fig. 4(a) to $[\text{Ho}^{3+}] = 0$, which gave $\tau_d = 100 \mu\text{s}$. Assuming that for very low Ho^{3+} concentration, W_t is negligible and $\tau_R = 1.12$ ms (Ref. 11) we obtain a value for the nonradiative decay rate, $W_{\text{nr}} = 9107 s^{-1}$ for the 5I_6 level. The following values of W_t were obtained: (i) $76.6 s^{-1}$ ($[\text{Ho}^{3+}] = 0.5\%$), (ii) $379 s^{-1}$ ($[\text{Ho}^{3+}] = 1\%$), (iii) $445 s^{-1}$

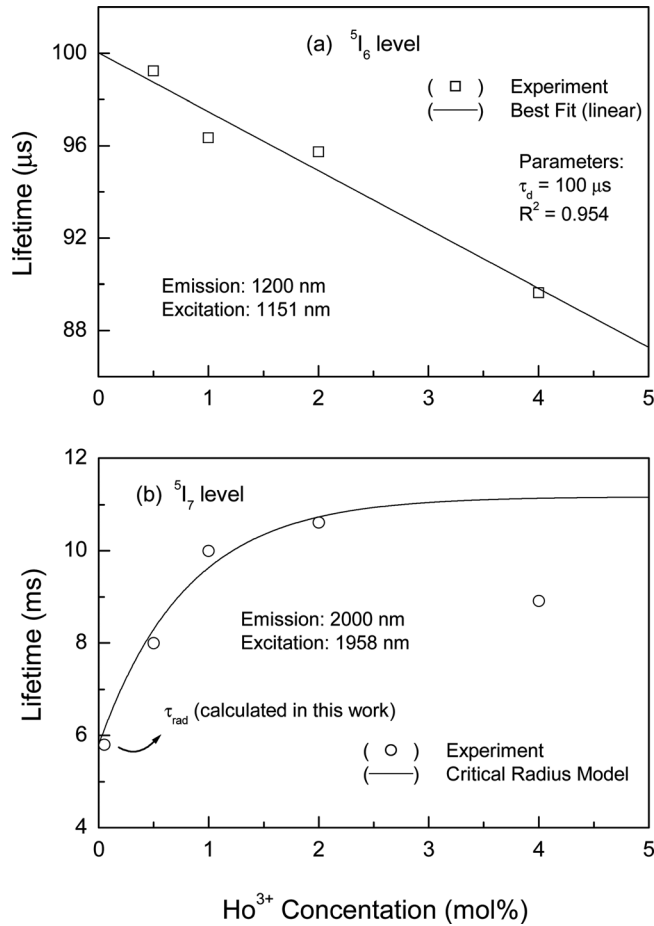


FIG. 4. Measured decay time (τ) of (a) ${}^5\text{I}_6$ level measured at 1200 nm using laser excitation at 1151 nm ($E = 10$ mJ, 4 ns) and (b) ${}^5\text{I}_7$ level measured for the luminescence decay at 2000 nm using the laser excitation at 1958 nm ($E = 10$ mJ, 4 ns) for Ho^{3+} -doped TZBG glass as a function of $[\text{Ho}^{3+}]$. The solid line in (b) represents the best fit using the critical radius model given by Eq. (5), where $\tau_{\text{rad}} = 5.8$ ms, and the fitting parameters are $\tau_0 = 5.3$ ms and $N_C = 0.8$ mol. %.

($[\text{Ho}^{3+}] = 2\%$), and (iv) 1157 s^{-1} ($[\text{Ho}^{3+}] = 4\%$). These values for W_t are essentially linearly dependent on $[\text{Ho}^{3+}]$ and indicate that ${}^5\text{I}_6$ energy level migration is important. As a consequence of energy transfer to OH^- , the luminescence efficiency from ${}^5\text{I}_6$ level decreases from 8.9 to 8% for $[\text{Ho}^{3+}] = 4\%$. From these calculations it is clear that the main decay process for the ${}^5\text{I}_6$ level is multiphonon decay.

C. Emission decay from the ${}^5\text{I}_7$ level

Figure 5 shows the emission decay from the ${}^5\text{I}_7$ level measured at 2000 nm by pulsed laser excitation at 1958 nm ($E = 10$ mJ). The emission decay from the ${}^5\text{I}_7$ state is observed to be exponential (see broken lines in Fig. 5), which indicates weak energy transfer to free OH^- radicals. It can be observed from Fig. 4(b) that the ${}^5\text{I}_7$ decay time initially increases with increasing Ho^{3+} concentration up to $[\text{Ho}] \sim 2\%$ and displays a slight decrease to $[\text{Ho}] = 4\%$. The measured decay times for the ${}^5\text{I}_7$ level are longer than the calculated radiative lifetime ($\tau_{\text{R}} = 5.8$ ms, calculated above). The increase level in the decay time of the ${}^5\text{I}_7$ in our experiment relates to strong excitation migration from a nonradia-

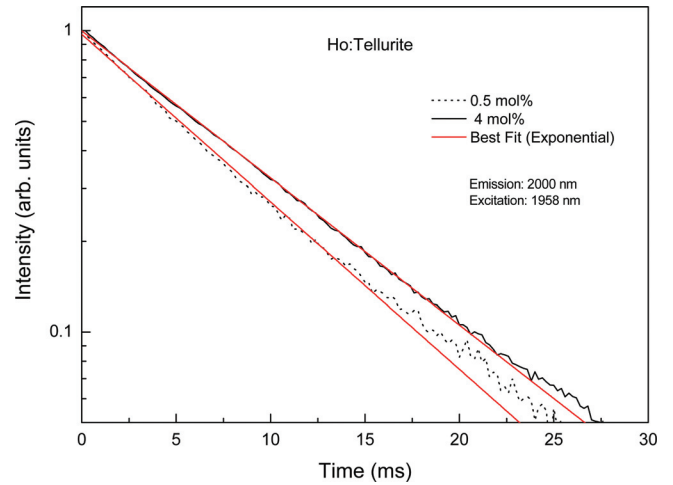


FIG. 5. (Color online) Measured emission decay characteristics (solid lines) of the ${}^5\text{I}_7$ level of Ho^{3+} -(0.5 and 4 mol. %)-doped TZBG glass using a short pulse laser excitation at 1958 nm with an average energy of 10 mJ and pulse duration of 4 ns at 10 Hz. Broken lines represent the best fit using an exponential decay model.

tive resonant dipole-dipole interaction between the ${}^5\text{I}_7$ and ${}^5\text{I}_8$ energy levels. This phenomenon explains the initial increase in the decay time as the Ho^{3+} concentration is increased. Note that measuring the luminescence close to the pumped surface can minimize radiation trapping. In our experiment, we observed an increase in the decay time at distances perpendicular to the pump of ~ 8 mm, however, for distances of ~ 3 mm and shorter (from where we took the measurements) the measured decay time was invariable.

The line of best fit to the decay time characteristic [open circles in Fig. 4(b)] with the increase in $[\text{Ho}^{3+}]$ was carried out using Eq. (5). Equation (5) predicted that the lifetime of the ${}^5\text{I}_7$ level increases if a Ho^{3+} ion excited to the ${}^5\text{I}_7$ level will transfer excitation to a ground state Ho^{3+} ion within a critical distance, R_C , with a decay time augmentation efficiency, $\eta(\tau)$, dependent on the Ho^{3+} concentration.^{15,16} This model predicts a saturation of the decay time increase to a value ($\tau_{\text{R}} + \tau_0$), which should be reached for $N_{\text{Ho}} > N_C$, where N_C is the critical concentration calculated from

$$\eta(\text{Ho}) = 1 - \exp(-N_{\text{Ho}}/N_C) \quad \tau_d = \tau_{\text{R}} + \eta(\text{Ho})\tau_0 \quad (5)$$

where $\tau_{\text{R}} = 5.8$ ms and N_{Ho} is the Ho^{3+} concentration. The best fit using Eq. (4) gives $\tau_0 = 5.3$ ms and $N_C = 0.8$ mol. % [see solid line in Fig. 4(b)].

No up-conversion processes (ESA or ETU) starting from the ${}^5\text{I}_7$ or ${}^5\text{I}_6$ levels were observed in our measurements. The expected green emission from the ${}^5\text{S}_2$ level or the red emission from the ${}^3\text{F}_5$ level was not observed despite the fact that it can be easily observed in Ho^{3+} -doped ZBLAN glass under pulsed laser excitation at 1958 nm (Ref. 16) or in Er^{3+} -doped tellurite glass with 980 nm pumping.²¹

IV. DISCUSSION

Tellurite glasses are an interesting alternative host material for shortwave and midwave infrared fiber lasers; because they combine a relatively low phonon energy with higher

thermal and chemical stability compared with, for example, fluoride glasses. On the other hand, tellurite glasses display lower luminescence efficiencies compared to fluoride glasses but they display higher emission cross sections and are easier to handle and splice to silica fibers. Tellurite glasses have shown efficient emission at $\sim 2 \mu\text{m}$ from Tm^{3+} , Ho^{3+} -codoped tellurite glass optical fiber lasers;^{17–19} however, there has been no demonstration of laser operation beyond $\sim 2 \mu\text{m}$ in tellurite glass.

The results presented here indicate that perhaps the main issue with these glasses are water incorporation and the low luminescence efficiency of the $^5\text{I}_6$ level. To illustrate this we have carried out a numerical simulation to predict the performance of Ho^{3+} -doped tellurite fiber lasers. The experimental observation that ETU1 is weak in Ho^{3+} -doped TZBG glass (at least up to $[\text{Ho}^{3+}] = 4 \text{ mol. \%}$) implies that unlike the Er^{3+} -doped ZBLAN system,²⁰ energy recycling is not possible for moderately concentrated Ho^{3+} -doped tellurite glasses when pumped into the $^5\text{I}_6$ level.

The rate equations were written to include the essential energy levels and pump and decay processes for laser emission at 2930 nm when the $^5\text{I}_6$ level is excited with pump radiation. The rate equations are

$$\frac{dn_1}{dt} = -R_P n_1 + \frac{n_2}{\tau_2} + \frac{\beta_{31}}{\tau_{R3}} n_3, \quad (6)$$

$$\frac{dn_2}{dt} = -\frac{n_2}{\tau_2} + \frac{\beta_{32}}{\tau_{R3}} n_3 + W_{nR}(32)n_3 + W_t(2)n_3, \quad (7)$$

$$\frac{dn_3}{dt} = R_P n_1 - \frac{n_3}{\tau_{R3}} - W_{nR}(32)n_3 - W_t(2)n_3, \quad (8)$$

where n_1 , n_2 , and n_3 indicate the $^5\text{I}_8$, $^5\text{I}_7$, and $^5\text{I}_6$ levels and $n_1 + n_2 + n_3 = x$, where $x = 0.005, 0.01, 0.02$, and 0.04 (i.e., mole fraction). The radiative lifetimes, luminescence branching ratios, and nonradiative multiphonon decay rate values that were obtained from our experiment are: $\tau_{R3} = 1.12 \text{ ms}$, $\beta_{32} = 0.16$, $\beta_{31} = 0.84$, $\tau_2 = 8.9 \text{ ms}$ (4 mol. %), $W_{nR}(32) = 9107 \text{ s}^{-1}$, and $W_t(2) = 1157 \text{ s}^{-1}$. Note that $W_{nR}(21) \sim 0$. The pump rate (R_P) can be converted to pump irradiance (I_P) using the relation: $R_P(\text{s}^{-1}) = \frac{\sigma_{\text{abs}}(\lambda)}{h\nu} I_P(\text{Wcm}^{-2})$. Using the absorption cross section $\sigma_{\text{abs}}(\lambda) = 3.3 \times 10^{-21} \text{ cm}^2$ for the pump wavelength $\lambda = 1153 \text{ nm}$, one obtains $R_P = 1000 \text{ s}^{-1}$ for a pump intensity $I_P = 52.2 \text{ kW cm}^{-2}$.

The calculated evolution of the excited state populations n_3 and n_2 (in cm^{-3}) were used to determine the population difference ($n_3 - n_2$), as shown in Fig. 6. One can see that steady state is reached in a time $< 2 \text{ ms}$. At equilibrium, the populations n_1 , n_2 , and n_3 were used to calculate the population inversion $\Delta n = n_3 - n_2$ for $[\text{Ho}^{3+}] = 0.5, 1, 2$, and 4 mol. \% for a number of pump intensities at 1153 nm. For all the simulations, we obtained a steady state but negative population inversion [even with $W_t(2) = 0$]. A positive population inversion, however, occurs for pump times shorter than $100 \mu\text{s}$ with a maximum in Δn occurring $55 \mu\text{s}$ after the pump is switched on; see Fig. 7(a). This result indicates that maximum laser emission can be achieved by pump pulses of $55 \mu\text{s}$ duration with a minimal interval of time between pulses of 50 ms . [This time interval is necessary for complete

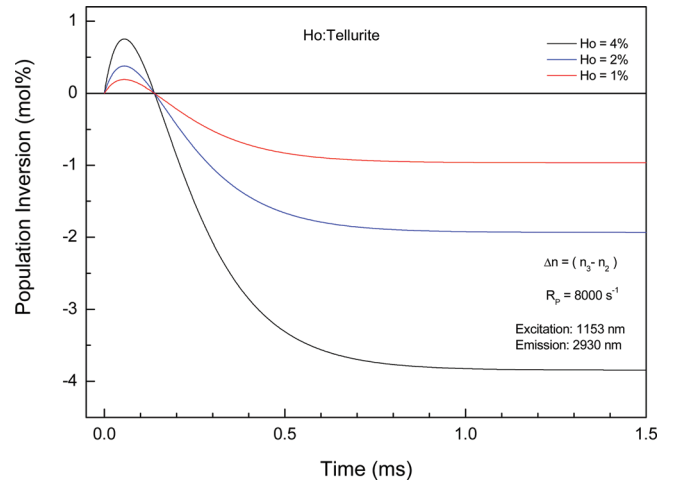


FIG. 6. (Color online) Calculated population inversion ($n_3 - n_2$) for cw pumping of Ho^{3+} -doped TZBG glass using a pumping intensity of $\sim 418 \text{ kW cm}^{-2}$ ($R_P = 8000 \text{ s}^{-1}$) for $[\text{Ho}^{3+}] = 1, 2$, and 4 mol. \% .

emptying of the $^5\text{I}_7$ level, as shown in Fig. 7(b)]. Using a square modulated $50 \mu\text{s}$ duration pump one can get $\Delta n = 0.75 \text{ mol. \%}$ for $[\text{Ho}^{3+}] = 4 \text{ mol. \%}$.

Figure 8 shows the maximum calculated population inversion using square 1153 nm pump pulses of varying

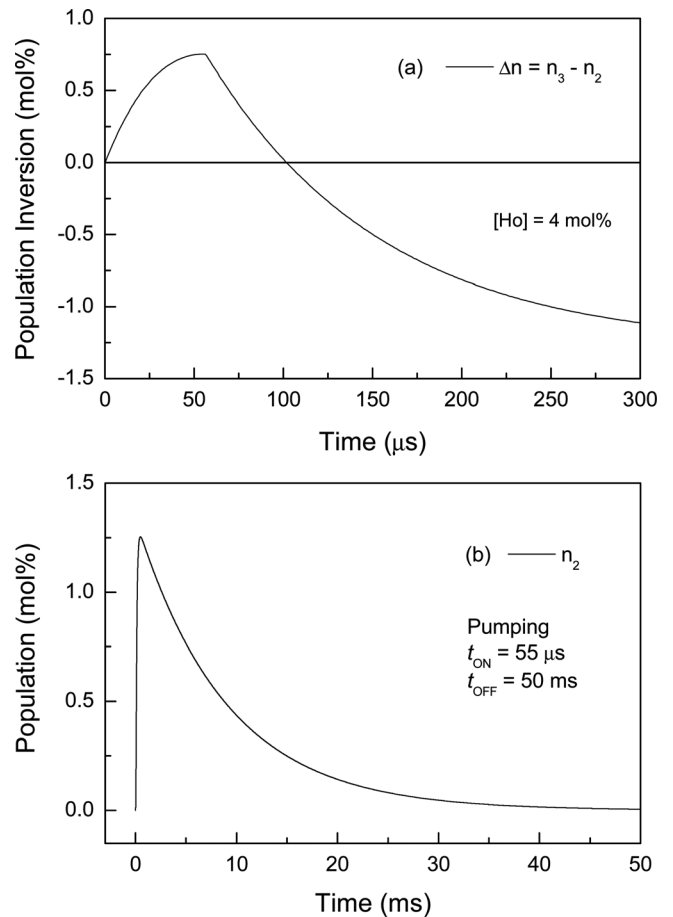


FIG. 7. Calculated (a) population inversion ($n_3 - n_2$) for a square pump pulse of $55 \mu\text{s}$ duration and pulse period = 50 ms and (b) $^5\text{I}_7$ population (n_2). The pump intensity was 418 kW cm^{-2} at 1153 nm , $[\text{Ho}^{3+}] = 4 \text{ mol. \%}$, $W_t = 0$.

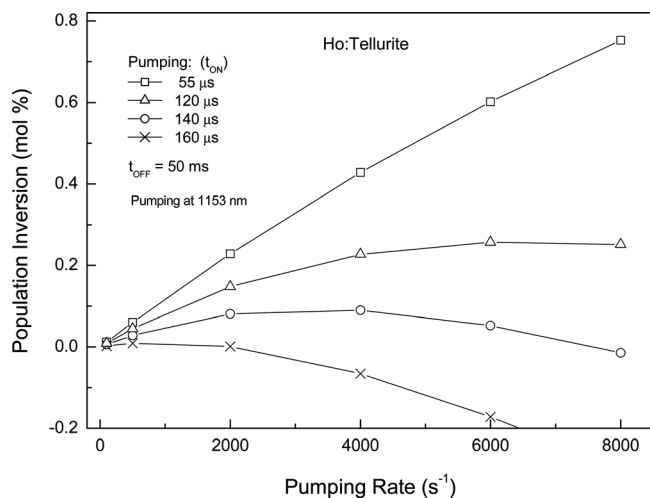


FIG. 8. Calculated population inversion for a square pump pulse for several pulse durations and a fixed interpulse period of 50 ms for Ho³⁺ (4 mol. %) doped TZBG glass as a function of the pump rate (R_P).

duration for [Ho³⁺] = 4 mol. % as a function of the pump rate (R_P). It can be observed that the population inversion is greatest when $t_{ON} = 55 \mu\text{s}$ and the population inversion decreases with increasing pumping duration. Note that for these simulations, no losses were included and, hence, the simulations are only a guide to possible performance from a fiber laser arrangement.

V. SUMMARY AND CONCLUSIONS

The decay processes relating to the $^5I_7 \rightarrow ^5I_8 \sim 2 \mu\text{m}$ and $^5I_6 \rightarrow ^5I_7 \sim 2.9 \mu\text{m}$ laser transitions in singly Ho³⁺-doped tellurite (TZBG) glass were investigated in detail using selective laser excitation of the 5I_6 energy level at 1151 nm and 5I_7 energy level at 1958 nm. The results have established that a nonradiative ETU between two excited Ho³⁺ ions in the 5I_7 level is negligibly weak for Ho³⁺ concentrations ≤ 4 mol. %. Pump ESA from the 5I_7 or 5I_6 levels was not observed when these levels were excited directly. The 5I_7 and 5I_6 energy levels were measured to emit luminescence with peaks at ~ 2050 nm and ~ 2930 nm, respectively. The 5I_6 level has low luminescence efficiency of 8% because of nonradiative multiphonon relaxation, however, decay from the 5I_7 level was observed to be fully radiative. The decreasing decay time of the 5I_6 level with increasing [Ho³⁺] was attributed to energy transfer to the OH⁻ radicals present

in the glass. The numerical simulations indicate that the prospect for cw operation on the $^5I_6 \rightarrow ^5I_7$ transition is low in Ho³⁺-doped tellurite glass.

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